Dynamical heterogeneity and structural relaxation in periodically deformed polymer glasses

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Movies, preprints @ http://www.wright.edu/~nikolai.priezjev/



N. V. Priezjev, "Dynamical heterogeneity in periodically deformed polymer glasses", *Physical Review E* **89**, 012601 (2014).

N. V. Priezjev, "Heterogeneous relaxation dynamics in amorphous materials under cyclic loading", *Physical Review E* **87**, 052302 (2013).

Dynamical heterogeneity in quiescent and deformed polymer glasses

Molecular dynamics simulations of a <u>quiescent</u> polymer glass near T_g



The most mobile monomers (top 5%) form transient clusters whose mean size increases upon cooling towards T_g

Gebremichael, Schroder, Starr, Glotzer, J. Chem. Phys. 64, 051503 (2001).



Riggleman, Lee, Ediger, and de Pablo, Soft Matter **6**, 287 (2010).

Dynamical heterogeneity in granular media and supercooled liquids



Spatial location of successive clusters of cage jumps Power-law distribution of clusters sizes Candelier, Dauchot, and Biroli, *PRL* **102**, 088001 (2009).

• <u>Fluidized bed experiment: Monolayer of bidisperse beads</u> Candelier, Dauchot, and Biroli, *EPL* **92**, 24003 (2010).

Present study:

- 3D polymer glasses under periodic strain?
- Monomer diffusion depends on the strain amplitude
- Structural relaxation and dynamical heterogeneities
- Particle hopping dynamics clusters of mobile particles
- Dynamical facilitation of mobile monomers

• <u>2D softly repulsing particle molecular dynamics simulation (supercooled liquids at $\gamma=0$)</u> Candelier, Widmer-Cooper, Kummerfeld, Dauchot, Biroli, Harrowell, Reichman, *PRL* (2010).

Details of molecular dynamics simulations and parameter values

Lennard-Jones
potential:
$$V_{LJ}(r) = 4\varepsilon \left[\left(\frac{r}{\sigma} \right)^{-12} - \left(\frac{r}{\sigma} \right)^{-6} \right]$$

FENE bead-
spring model: $V_{FENE}(r) = \frac{1}{2} k r_o^2 \ln \left(1 - \frac{r^2}{r_o^2} \right)$
 $k = 30\varepsilon\sigma^{-2}$ and $r_o = 1.5\sigma$

Kremer & Grest, J. Chem. Phys. 92, 5057 (1990)

Monomer density: $\rho = 1.07 \,\sigma^{-3}$, N = 3120Temperature: $T = 0.1 \,\varepsilon/k_B < T_g \approx 0.3 \,\varepsilon/k_B$ Cubic box: $14.29 \,\sigma \times 14.29 \,\sigma \times 14.29 \,\sigma$ Lees-Edwards periodic boundary conditions SLLOD equations of motion: $\Delta t_{MD} = 0.005 \,\tau$

Oscillation period: $T = 2\pi / \omega = 125.66 \tau$

Oscillatory shear strain: $\gamma(t) = \gamma_0 \sin(\omega t)$ Strain amplitude: $\gamma_0 \le 0.09$, $\omega = 0.05\tau^{-1}$



15,000 cycles ($\approx 3.8 \times 10^8$ MD steps)

Mean-square-displacement of monomers for different strain amplitudes γ_0



Time autocorrelation function of normal modes vs. strain amplitude γ_0

$$C_{p}(t) = \left\langle X_{p}(t) \cdot X_{p}(0) \right\rangle \qquad X_{p}(t) = \frac{1}{N} \sum_{i=1}^{N} r_{i}(t) \cos \frac{p\pi(i-1/2)}{N} \qquad p = \text{mode number} \\ N = 10 \text{ chain length} \\ \hline N = 0.05 \\ \hline N = 0.07 \\ \hline N = 0.05 \\ \hline N = 0.07 \\ \hline N = 0.05 \\ \hline N = 0.07 \\ \hline N = 0.05 \\ \hline N = 0$$

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Oscillation period: $T = 2\pi / \omega = 125.66 \tau$

Self-overlap order parameter $Q_s(a,t)$ for different strain amplitudes γ_0

$$Q_s(a,t) = \frac{1}{N_m} \sum_{i=1}^{N_m} \exp\left(-\frac{\Delta r_i(t)^2}{2a^2}\right)$$

 $a = 0.12\sigma$ = probed length scale (about the cage size) N_m = total number of monomers in the system $\Delta r_i(t)$ = the monomer displacement vector



 $Q_s(a,t)$ describes structural relaxation of the material. Measure of the spatial overlap between monomer positions.

<u>Reversible dynamics</u>: $Q_s(t) \approx \text{constant}$

<u>Diffusive regime</u>: $Q_s(t)$ vanishes at large t

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Susceptibility $X_4(a,t) = ?$

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Contour plots of dynamical susceptibility $X_4(a,t)$ and $Q_s(a,t)$ for $\gamma_0 = 0.06$



$$Q_s(a,t) = \frac{1}{N_m} \sum_{i=1}^{N_m} \exp\left(-\frac{\Delta r_i(t)^2}{2a^2}\right)$$

Measure of the spatial overlap between monomer positions.

$$X_4(a,t) = N_m \cdot \left(\left\langle Q_s(a,t)^2 \right\rangle - \left\langle Q_s(a,t) \right\rangle^2 \right)$$

 $X_4(a,t)$ is dynamical susceptibility, which is the variance of Q_s .

Maximum $X_4(a,t)$ indicates the largest number of monomers involved in correlated motion.

Berthier & Biroli (2011)

Dynamical susceptibility as a function of time for different strain amplitudes γ_0

$$X_4(a,t) = N_m \left(\left\langle Q_s(a,t)^2 \right\rangle - \left\langle Q_s(a,t) \right\rangle^2 \right)$$



 N_m = total number of monomers $X_4(a,t)$ is dynamical susceptibility, which

probed length scale $a = \max \operatorname{in} X_4(a, t)$

Maximum $X_4(a,t)$ indicates the largest spatial correlation between localized monomers.

Berthier & Biroli (2011)

Transition at $\gamma_0 = 0.06$ maximum correlation length $\xi_4 = [X_4^{\max}(a,t)]^{1/3}$

(in contrast to steadily sheared supercooled liquids and glasses) Mizuno & Yamamoto, JCP (2012), Tsamados, EPJE (2010)

Number of monomers undergoing cage jumps N_c as a function of time t/T



Numerical algorithm for detection of cage jumps:



Candelier, Dauchot, Biroli, *PRL* (2009).

Periodic deformation = intermittent bursts of large monomer displacements.

Scale-invariant processes or Pink noise = "1/f noise"

Power spectrum ~ frequency⁻² = simple Brownian noise

Typical clusters of mobile monomers for different strain amplitudes γ_0



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Fraction of dynamically facilitated monomers increases with strain amplitude



Oscillation period: $T = 2\pi / \omega = 125.66 \tau$

Vogel and Glotzer, *Phys. Rev. Lett.* **92**, 255901 (2004).

Large surface area = high probability to have mobile neighbors.

Important conclusions:



- The coarse-grained bead-spring polymer glass at $T = 0.1 \epsilon / k_B < T_g$ under spatially homogeneous, time-periodic shear strain.
- At small strain amplitudes, the mean square displacement exhibits a broad sub-diffusive plateau and the system undergoes nearly <u>reversible deformation</u> over about 10⁴ cycles.
- At the critical strain amplitude, the transition from slow to fast relaxation dynamics is associated with the largest number of <u>dynamically correlated monomers</u> as indicated by the peak value of the dynamical susceptibility. $\xi_4 = [X_4^{\text{max}}(a,t)]^{1/3}$
- The detailed analysis of <u>monomer hopping dynamics</u> indicates that mobile monomers aggregate into clusters whose sizes increase at larger strain amplitudes. (This is in contrast to steadily sheared supercooled liquids and glasses).
- Fraction of <u>dynamically facilitated</u> mobile monomers increases at larger strain amplitudes.

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